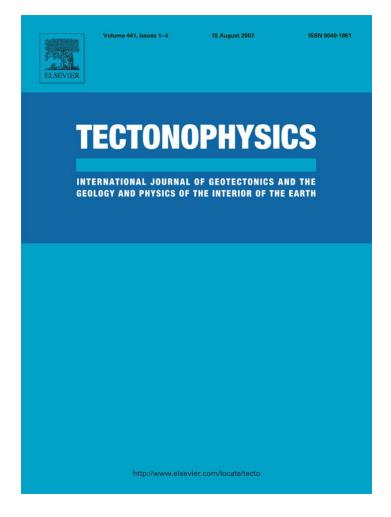
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Influence of time, temperature, confining pressure and fluid content on the experimental compaction of spherical grains

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Abstract

Theoretical models of compaction processes, such as for example intergranular pressure-solution (IPS), focus on deformation occurring at the contacts between spherical grains that constitute an aggregate. In order to investigate the applicability of such models, and to quantify the deformation of particles within an aggregate, isostatic experiments were performed in cold-sealed vessels on glass sphere aggregates at 200 MPa confining pressure and 350 °C with varying amounts of fluid. Several runs were performed in order to investigate the effects of time, fluid content, pressure and temperature, by varying one of these parameters and holding the others fixed. In order to compare the aggregates with natural materials, similar experiments were also performed using quartz sand instead of glass spheres. Experiments with quartz show evidence of IPS, but the strain could not be quantified. Experiments with glass spheres show evidence of several types of deformation processes: both brittle (fracturing) and ductile (plastic flow and fluid-enhanced deformation, such as IPS). In experiments with a large amount of water (≥5 vol.%), dissolution and recrystallization of the glass spheres also occurred, coupled with crystallization of new material filling the initial porosity. Experiments performed with a fluid content of less than 1 vol.% indicate creep behavior that is typical of glass deformation, following an exponential law. These experiments can also be made to fit a power law for creep, with a stress exponent of $n=10.5\pm$ 2.2 in both dry and wet experiments. However, the pre-factor of the power law creep increases 5 times with the addition of water, showing the strong effect of water on the deformation rate. These simple and low-cost experiments provide new insights on the rheology of soda-lime glass, which is used in analogue experiments, and of glass-bearing rocks under mid-crustal P-T conditions. They also highlight the strong enhancement of plasticity of natural rocks in presence of fluid or of a glassy phase. © 2007 Elsevier B.V. All rights reserved.

Keywords: Isostatic experiments; Glass spheres; Ductile deformation; Compaction

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1. Introduction

Ductile deformation usually occurs in the lower crust under high pressure (P) and temperature (T)conditions by crystal plasticity, but can also be efficient under mid-crustal P-T conditions when deformation is accommodated by diffusive mass transfer through a fluid phase (intergranular pressure-solution - IPS -; e.g., Weyl, 1959; Rutter, 1976; Tada and Siever, 1989; Gratier et al., 1999). IPS results from a stress gradient at grain-scale that leads to dissolution and flattening of grain contacts, transport through the contacts to the pores, and precipitation of new phases within the porosity (e.g., Weyl, 1959; Tada and Siever, 1989). At higher confining pressure, plastic deformation can occur at grain contacts leading to contact flattening. These two ductile deformation mechanisms thus compact and strengthen the initial aggregate, leading to a reduction in porosity and permeability.

Numerous experimental studies have been performed in order to understand the mechanical processes and to quantify the strain of mineral aggregates. High-temperature compaction experiments have been performed in order to investigate plastic deformation in the lower crust and in high-strain zones using both dry and hydrated aggregates (e.g., Rybacki and Dresen, 2000; Xiao et al., 2002; Rutter and Brodie, 2004). Under the P-Tconditions of the middle and upper crust, creep experiments were performed in the presence of fluid, including 1) compaction experiments in drained conditions, (e.g., Renton et al., 1969; Rutter, 1983; Kronenberg and Tullis, 1984; Gratier and Guiguet, 1986; Rutter and Wanten, 2000), 2) compaction experiments with a controlled fluid pressure (i.e. the effective pressure: Niemeijer et al., 2002; Niemeijer and Spiers, 2002; He et al., 2003), and 3) shearing experiments (Bos and Spiers, 2000; Stünitz and Tullis, 2001).

Supplementing the experimental approaches, several numerical and theoretical models have been proposed to explain creep compaction (e.g. Weyl, 1959; Rutter, 1976; Raj and Chyung, 1981; Tada and Siever, 1986; Tada et al., 1987). These models are generally based on the deformation of spherical elements making up an aggregate (e.g., Lemée and Guéguen, 1996; Renard et al., 1999, 2000; Gundersen et al., 2002; Yasuhara et al., 2003). The comparison between models, experiments and nature is not straightforward because compaction experiments are generally conducted with grains of irregular shape, and because models focus on grain-scale deformation, whereas compaction experiments focus on bulk deformation of the aggregate.

Considering an aggregate in a closed system, subjected to an external hydrostatic pressure, the effective normal stress at the grain contacts depends on the external confining pressure, the pore fluid pressure, and the contacts surface area that increases with compaction. Therefore, the final morphology of the spheres and the size of the contact areas can be used to estimate the local effective stress at the grain contacts at the end of experimental runs. Experiments of different run times involving calibrated spheres instead of grains of irregular shape, conducted in closed systems with different starting solid/fluid ratios could be used to quantify strain at the particle scale and to constrain general strain-stress relations, even without a proper knowledge of the applied stress or strain rate. The aim of the present study was to test this "grain-scale" approach by conducting compaction experiments on an aggregate of calibrated spherical particles, in order to quantify particle strain at the end of the runs by observing the grain contacts using a Scanning Electron Microscope (SEM).

Such experiments could be conducted using very simple experimental equipment, such as isostatic coldsealed autoclaves or internally-heated apparatus, thus reducing the duration of experiment preparation, and allowing experiments to be performed over a wide range of confining pressure and temperature values. The main disadvantage of such an experimental set-up is the change in effective stress during the experiment, due to an increase in the contact surface area between the spheres with increasing compaction. Therefore, the strain rate and the stress at the contacts are not controlled and vary during the experiment, making it difficult to derive general compaction laws. However, the combination of experiments conducted with the same initial conditions (sphere diameter, water/sphere ratio, temperature, and confining pressure) but different run durations should allow the time-dependent change in strain rate and local stress to be estimated at the contact of the spheres with deformation. To the authors' knowledge, such an approach has not been attempted so far as it requires working with aggregates with constant particle geometry and size. Owing to the difficulties in producing perfectly spherical mineral grains with a small and constant size of about 100 µm, which are required for such experiments, commercial glass spheres had to be used. Moreover, dissolution and precipitation kinetics of glass are higher than in crystallized materials, thus allowing shorter experimental durations.

The present work reports on experiments conducted in closed system conditions varying the time, the solid/ fluid ratio, the confining pressure and the temperature. The results open up the possibility of studying rock deformation under isostatic confining pressure, and using glass spheres as an analogous material for silicate minerals. For comparison with natural minerals, similar experiments were conducted with non-spherical quartz grains instead of glass spheres. The results obtained from this very simple experimental approach are used to discuss the creep law of an aggregate made of ductile glass beads and the controlling effect of water content, temperature and pressure on deformation.

2. Experimental method

The starting material was an aggregate of 45–90 μm diameter spheres of soda-lime glass (72% SiO₂, 14% Na₂O, 10% CaO and 3% MgO) from Sandmaster—France. At atmospheric pressure this glass has a soft-

ening point at 730 °C. According to Fig. 1, and assuming that viscosity increases with pressure, the glass should behave elastically under stress at T< 450 °C. Therefore, and since it is the IPS processes that were of prime interest, most experiments were conducted at 350 °C and 200 MPa confining pressure. For comparison with crystallized material, additional experiments were performed with quartz sand (Nemours sand; d_{50} =199 µm). In order to investigate the variation in strain with time, several runs were made at the same P-Tconditions for durations ranging from 2 h to 4 weeks. To investigate the effect of temperature at constant pressure, a set of experiments was conducted at 25, 150, 250, 350 and 450 °C and a constant pressure of 200 MPa. The effect of pressure was studied at a constant temperature of 350 °C, with experiments conducted at 100, 200, 300 and 400 MPa (Table 1).

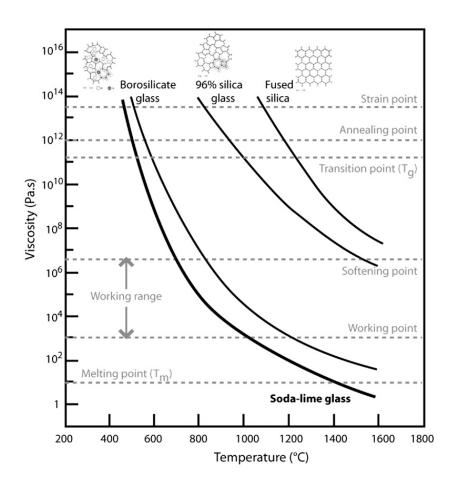


Fig. 1. Comparison of the viscosity of several kinds of glass versus temperature. Some characteristic temperatures are also reported: melting temperature ($T_{\rm m}$, η =10 Pa s) below which glass is a liquid; working range (η =10³-10⁴ Pa s) for which glass is easily deformed; softening temperature (η =4×10⁶ Pa s) which is the maximum temperature for which a glass sample does not flow under its own load; glass transition temperature ($T_{\rm g}$, η =2×10¹¹ Pa s) corresponds to the transition from supercooled liquid to solid glass; annealing temperature (η =10¹² Pa s) at which glass releases its internal stress (by solid diffusion); strain temperature (η =3×10¹³ Pa s) below which brittle deformation occurs before plastic deformation. Under the experimental conditions of the present study (T<450 °C) the viscosity of soda-lime glass used (bold curve) is higher than 10¹⁵ Pa s, thus indicating that elastic brittle deformation occurred in addition to viscous flow (modified from a document from a lecture by L.V. Zhigilei at University of Virginia: http://www.people.virginia.edu/~lz2n/mse209/Chapter14.pdf).

Table 1
Data of experiments performed with glass spheres and quartz grains

	Amount	of fluid a	P	T	t	d _c	Uncertainty	λī	o c	$\Delta \epsilon^{\rm b}$,	Δh^{b}	dε/dt	2∑ ^d	In do/dt	25 d	log dε/dt	25 d	_ e	$\Delta \sigma^{\rm b}$	log et	2∑ ^d	
	Amount	or mind	P	1	I	u _c	d_c^b	N _c	3	Δε	h	Δh^{-}		2Σ"	in ue/ui	22	log d€/dI	22	(MPa)		logσ	25.	(Pa s)
	(vol.%)	Deviation (%)	(MPa)	(°C)	(h)	(µm)	(µm)				(µm)	(µm)			(s ⁻¹)		(s ⁻¹)				(Pa)		
xperimer	ıt with glo	ass beads																					
mW16	1	-5	200	350	0	18.74	0.68	158	0.057	0.019	2.00	0.17	4.30×10^{-7}		-14.66	0.34	-6.37	0.15	3724	445	9.57	0.10	8.66×10^{15}
mW14	1	10	200	350	2	21.78	0.90	117	0.051	0.004	1.79	0.15	2.25×10^{-7}	8.04×10^{-8}	-15.31	0.36	-6.65	0.16	3102	284	9.49	0.08	1.38×10^{16}
mW12	1	17.5	200	350	6	21.45	0.76	139	0.051	0.004	1.79	0.14	1.18×10^{-7}	4.45×10^{-8}	-15.95	0.38	-6.93	0.16	3123	303	9.49	0.08	2.65×10^{16}
mW10	1	17.5	200	350	24	22.38	0.90	97	0.050	0.004	1.75	0.15	3.93×10^{-8}	1.61×10^{-8}	-17.05	0.41	-7.41	0.18	3528	1033	9.55	0.25	8.98×10^{16}
mW3	1	12.5	200	350	168	24.39	0.81	80	0.058	0.005	2.03	0.15	6.82×10^{-9}	3.16×10^{-9}	-18.80	0.46	-8.17	0.20	2438	196	9.39	0.07	3.57×10^{17}
mW2	1	15	200	350	336	21.40	1.01	74	0.058	0.009	2.03	0.22	3.59×10^{-9}	1.74×10^{-9}	-19.45	0.48	-8.44	0.21	2951	435	9.47	0.13	8.22×10^{17}
mW8	1	-5	200	350	504	23.03	1.13	88	0.071	0.009	2.49	0.23	2.48×10^{-9}	1.23×10^{-9}	-19.82	0.49	-8.61	0.21	2530	515	9.40	0.18	1.02×10^{18}
mW5	1	-40	200	350	672	25.18	1.94	34	0.069	0.012	2.42	0.41	1.89×10^{-9}	9.51×10^{-10}	-20.09	0.50	-8.72	0.22	2485	592	9.40	0.21	1.31×10^{18}
mW15	5	3	200	350	2	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_	-	_
mW13	5	7.5	200	350	6	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_	-	_
mW11	5	-2	200	350	24	_	_	_	_	_	_	_	_	_	_	_	_	_	-	_	_	_	_
mW4	5	_	200	350	168	_	_	_	_	_	_	_	_	_	_	_	_	_	-	_	_	_	_
mW1	5	10.5	200	350	336	_	_	_	_	_	_	_	_	_	_	_	_	_	-	_	_	_	_
mW9	5	14.5	200	350	504	_	_	_	_	_	_	_	_	_	_	_	_	_	-	_	_	_	_
mW6	5	6.5	200	350	672	_	_	_	_	_	_	_	_	_	_	_	_	_	-	_	_	_	_
m6	Dry	_	200	350	1	16.19	0.41	217	0.032	0.002	1.12	0.06	3.36×10^{-7}	1.14×10^{-7}	-14.91	0.34	-6.47	0.15	4618	318	9.66	0.06	1.37×10^{16}
m10	Dry	_	200	350	2	18.18	0.96	71	0.037	0.004	1.30	0.05	1.76×10^{-7}	6.29×10^{-8}	-15.55	0.36	-6.75	0.16	3938	177	9.60	0.04	2.24×10^{16}
m5	Dry	_	200	350	2	16.57	0.37	242	0.032	0.002	1.12	0.13	1.76×10^{-7}	6.29×10^{-8}	-15.55	0.36	-6.75	0.16	4319	646	9.64	0.13	
m3	Drv	_	200	350	6	19.53	0.68	126	0.043	0.003	1.51	0.10	9.21×10^{-8}		-16.20		-7.04	0.16	3439	266	9.54	0.07	3.73×10^{17}
m1	Dry	_	200	350	24	21.03	1.42	72	0.043	0.005	1.51	0.20	3.07×10^{-8}	1.26×10^{-8}	-17.30	0.41	-7.51	0.18	3820	604	9.58	0.14	1.24×10 ¹⁷
m4	Drv	_	200	350	168	20.60	0.94	102	0.053	0.005	1.86	0.17	5.32×10^{-9}	2.47×10^{-9}	-19.05	0.46	-8.27	0.20	2963	431	9.47	0.13	5.5710^{18}
m2	Drv	_	200	350	336	21.04	0.66	177	0.049	0.003	1.72	0.11	2.80×10^{-9}	1.35×10^{-9}	-19.69	0.48	-8.55	0.21	3205	300	9.51	0.08	1.14×10^{18}
mW30	1	-5	200	150	1	16.30	0.73	66	0.027	0.003	0.95	0.10	_	_	_	_	_	_	5550	646	_	_	_
mW29	1	0	200	150	2	16.09	0.48	112	0.027	0.001	0.95	0.06	1.14×10^{-7}	4.07×10^{-8}	-15.99	0.36	-6.94	0.16	5338	374	9.73	0.06	4.68×10^{16}
mW17		12.5	200	150	6	15.39	0.58	135	0.024	0.001	0.84	0.06	5.99×10^{-8}	2.26×10^{-8}	-16.63		-7.22	0.16	6310	529	9.80	0.07	1.05 × 10 ¹
mW25		17.5	200	150	24	16.06	0.67	89	0.025	0.002	0.88	0.08	2.00×10^{-8}	8.21×10 ⁻⁹	-17.73		-7.70	0.18	6264	768	9.80		3.13×10 ¹⁷
mW18		15	200	150	168	16.69	0.81	80	0.029	0.004	1.02	0.14	3.46×10^{-9}	1.60×10^{-9}	-19.48		-8.46	0.20	5735	813	9.76	0.12	1.66×10 ¹⁸
mW26		45	200	150	336	-	-	_	-	-	- 1.02	_	_	_	_ 17.70	-	- 0.40	-	-	-			_
mW20	1	22.5	200	150	672	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_	_
m18	Dry	_	200	150	1	15.20	0.77	45	0.025	0.003	0.88	0.10	2.33×10^{-7}	7.87×10^{-8}	-15.27	0.34	-6.63	0.15	5868	712	9.77	0.11	2.5210 ¹⁶
m17	Dry	_	200	150	2		0.70	74		0.003	1.02	0.10	1.22×10^{-7}		-15.27		-6.91	0.15	5046	473	9.70		4.14×10 ¹⁶
1111 /	Diy	_	200	130	-	10.4/	0.70	/+	0.029	0.003	1.02	0.09	1.22 ^ 10	4.50 ^ 10	13.92	0.30	0.71	0.10	2040	T/3	2.70	0.08	7.14^

Gm7	Dry	_	200	150	6	14.93	0.68	95	0.024	0.003	0.84	0.09	6.41×10^{-8}	2.42×10^{-8}	-16.56	0.38	-7.19	0.16	6615	873	9.82	0.11	1.03×10^{16}
Gm13	Dry	-	200	150	24	17.04	0.72	59	0.031	0.003	1.09	0.09	2.13×10^{-8}	8.74×10^{-9}	-17.66	0.41	-7.67	0.18	4621	419	9.66	0.08	2.17×10^{17}
Gm8	Dry	-	200	150	168	14.97	0.69	97	0.025	0.002	0.88	0.08	3.70×10^{-9}	1.72×10^{-9}	-19.41	0.46	-8.43	0.20	6311	703	9.80	0.10	1.71×10^{18}
Gm14	Dry	-	200	150	336	_	_	-	-	_	_	_	_	_	_	_	_	_	-	_	-	-	-
GmW22	5	11.5	200	150	6	_	_	-	-	_	_	_	_	_	_	_	_	_	-	_	-	-	-
GmW21	5	-	200	150	6	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
GmW32	1	27.5	200	25	6	16.30	2.41	16	0.019	0.005	0.67	0.18	_	_	_	_	_	_	8216	2092	-	-	-
Gm20	Dry	-	200	25	6	15.51	1.10	26	0.025	0.004	0.88	0.14	_	_	_	_	_	_	5978	1132	-	-	-
GmW31	1	30	200	250	6	17.03	0.68	104	0.032	0.003	1.12	0.11	_	_	_	_	_	_	4611	445	-	-	-
Gm19	Dry	_	200	250	6	17.38	0.59	137	0.033	0.003	1.16	0.09	_	_	_	_	_	_	4590	388	-	-	-
GmW23	1	5	200	450	6	_	_	-	-	_	_	_	_	_	_	_	_	_	-	_	-	-	-
Gm11	Dry	_	200	450	6	_	_	-	-	_	_	_	_	_	_	_	_	_	3105	502	-	-	-
GmW28	1	12.5	100	350	6	18.90	0.45	218	0.043	0.003	1.51	0.08	_	_	_	-	_	-	1791	133	_	_	_
Gm16	Dry	_	100	350	6	16.61	0.61	105	0.030	0.003	1.05	0.08	_	_	_	_	_	_	2547	247	-	-	-
GmW27	1	-32.5	300	350	6	21.63	0.69	165	0.052	0.004	1.82	0.14	_	_	_	_	_	_	4586	494	-	-	-
Gm15	Dry	_	300	350	6	21.93	0.78	93	0.049	0.004	1.72	0.13	_	_	_	_	_	_	4441	463	-	-	-
GmW24	1	-5	400	350	6	22.87	1.25	81	0.063	0.013	2.21	0.46	_	_	_	_	_	_	5251	1365	-	-	-
Gm12	Dry	_	400	350	6	22.25	0.81	165	0.052	0.004	1.82	0.14	_	_	_	-	_	-	6256	681	_	_	_
ENS2	15	-4.5	260	350	816	_	_	-	-	_	_	_	_	_	_	_	_	_	-	_	-	-	-
Experime	nts with q	juartz grain:	S																				
QmW1	1	-10	200	350	168	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW2	5	4.5	200	350	168	_	_	-	-	_	-	_	_	_	-	-	_	-	_	_	_	_	_
QmW4	1	9	200	350	336	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW5	5	-5	200	350	336	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW6	1	2.5	200	350	1344	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW7	5	2.5	200	350	1344	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW8	1	-7.5	200	350	672	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW9	5	1	200	350	672	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW10	1	20	200	350	1008	-	_	-	-	-	-	-	_	-	-	-	-	-	-	-	-	-	-
QmW11	5	3	200	350	1008	-	-	-	-	-	-	-	-	-	-	-	-	-	-	-	-	-	-

P: Pressure; T: temperature; t: duration of each experiment; d_c : contact diameter; N_c : number of measured contact; ϵ : local 1-D strain; h: change in diameter due to the strain; $d\epsilon/dt$: strain—rate; σ : local stress; $\eta_{\rm eff}$: effective viscosity defined by $\eta_{\rm eff} = s/(d\epsilon/dt)$. Strain data are calculated only in experiments where a significant number of flattened grain contacts were detected on the SEM

a 1 vol.% fluid=0.4 μ L and 5 vol.% fluid=2 μ L. Deviations indicate the per cent deviation in the real amount of fluid apart from these values. b Uncertainties Δ are calculated as $2\sum/\sqrt{N_c}$ with \sum the standard deviation.

⁶ Uncertainties Δ are carculated as $2 \angle J$ ^c The strain is estimated using Eq. (1). ^d \sum is the standard deviation. ^e The stress is estimated using Eq. (3).

In all experiments, the glass spheres (or quartz grains) were mixed with about 3 wt.% biotite to investigate the possible enhancement of IPS at the contacts with phyllosilicates (Bos and Spiers, 2000; Rutter and Wanten, 2000; Niemeijer and Spiers, 2002). 100 mg of this was loaded with varying but controlled amounts of fluid into a 3.0 mm outside diameter, 0.2 mm wall thickness and 12 mm long cylindrical gold capsule (Fig. 2A). The confining pressure minus the pressure of the water in the capsule equals the effective confining pressure. To investigate the effect of fluid availability on deformation, various amounts of fluid were thus introduced in the sample: 0, 1 and 5 vol.% double de-ionized water (volume of fluid related to 100% volume of glass spheres). These amounts of water give rise to different fluid (P_f) and effective (P_g) pressures. Assuming a bodycentered cubic packing, the starting porosity of the sphere aggregate is about 32%. Considering an initial porosity of 32%, the density of the air+water fluid in the 5 vol.% water experiments was 0.12 g cm⁻³. From the P-V-T data of Burnham et al. (1969) and Keenan et al. (1969), this corresponds to a $P_f \sim 25$ MPa at 450 °C. The 0.12 g cm⁻³ iso-density curve intersects the vapor liquid equilibrium curve at about 350 °C and 17 MPa. Below this temperature, $P_{\rm f}$ lies on the liquid-vapor curve, i.e. about 10 MPa at 300 °C and 4 MPa at 150 °C. Experiments performed with 20 vol.% of fluid at 400 °C would thus have a fluid pressure of about 40 MPa. Therefore, the fluid pressure in the 1 to 5 water vol.% experiments was about 12% of the confining pressure $(P_{\rm c})$, and $P_{\rm f} < 0.1 \times P_{\rm c}$ in experiments conducted at T < 400 °C. The fluid pressure was higher in experiments conducted with higher amounts of fluid, but it was always less than $0.25 \times P_{\rm c}$ (maximum value calculated for the 400 °C experiment performed with 20 vol.% fluid). These calculations show that the effective pressure was almost the same or close to the confining pressure in most experiments.

Gold capsules were used because gold is not reactive to the materials and fluids used for this study. Moreover, gold deforms easily and the external confining pressure is completely transmitted to the enclosed experimental mixture. Once filled, the capsules were welded shut and placed in a water-pressurized horizontal cold-sealed pressure vessel (Tuttle, 1969; Fig. 2B). The capsules were weighed after each stage of sample preparation to check for any leakage. Confining pressure was measured to within ±5 MPa, and temperature was measured to within ±1 °C with a chromel-alumel thermocouple located inside the vessel at the contact with the gold capsule. Because of the strong inertia of the heating system and the small size of the capsule, the temperature gradient at the hot end of the autoclave and along the capsule was negligible (Vidal, 1997; Vidal and Durin, 1999).

In each experiment, pressure was applied first, and the sample was then heated up to the desired temperature in about 2 h, keeping the pressure constant. At the end of the runs, temperature was decreased

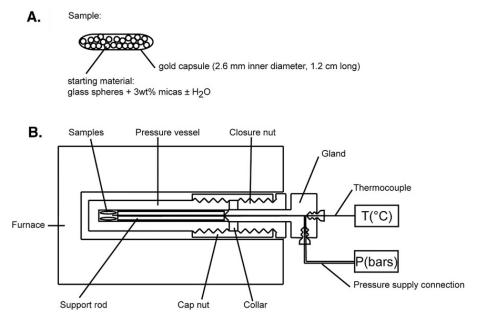


Fig. 2. Experimental set-up: A) Sample of glass spheres and fluid with controlled composition encapsulated within a 2 mm internal diameter and 20 mm long gold capsule (thickness of the walls: 0.2 mm). The confining pressure medium was water. B) The cold pressure vessel is located within a horizontal heating system. The confining pressure is controlled with water.

to 25 °C in less than 15 min. The remaining pressure was released when the samples had reached room temperature.

After the runs, the capsules were weighed to check for any water loss and leakage. The capsules were then opened and some material was collected for SEM observation. Except for experiments performed with ≥5 vol.% fluid, aggregate cohesion was lost during this procedure, which facilitated observations of the grain contact size and morphology. The run material was deposited on gold-coated copper holders and observed with a Zeiss DSM 962 SEM (GFZ Potsdam, Germany) and a Leica Stereoscan 440 SEM (Université de Savoie, Chambéry, France). Local strain at the grain contacts was calculated from the change in grain geometry measured on the SEM pictures (see Section 3.2). The amount of fluid that possibly entered the glass spheres during the experiments was estimated qualitatively from Raman spectroscopy, using a Jobin-Yvon LabRam HR800 (ENS-Lyon, France), conducted directly on the spheres or on double polished thin sections made from epoxyimpregnated samples.

3. Results

3.1. Effect of water content on the morphology of contacts

The morphology of the grain-grain contacts characterized under the SEM is strongly dependent on the amount of fluid within the sample.

In dry experiments, the contacts between the grains showed flat circular morphologies (Fig. 3A) and that the spheres were not bound together. Despite flattening at the contact, there was no significant reduction in porosity after the runs. In experiments with 1 vol.% fluid, most spheres were bound together and the contacts showed a flat rim and conchoidal cracking at the core (Fig. 3B). Numerous contacts showed concentric circles of alternating outgrowths and depressions (Fig. 3C), and rare stylolite-like contacts were observed, with the spheres both dissolving and penetrating one into another (Fig. 3D). In both dry and 1 vol.% fluid experiments, the initial porosity was well maintained despite deformation: porosity reduction was too small to be observed under the SEM (Fig. 3A-B). On the contrary, the initial porosity in experiments conducted with ≥ 5 vol.% fluid was completely filled with a microporous matrix having the same composition as the glass spheres. The spheres were all bound together, and their free faces showed recrystallized rims (Fig. 3E-F). Porosity filling and sphere bonding

combined to strengthen the material. The sphere diameter was slightly smaller than that of the starting material, and sphere interpenetration was observed, both these phenomena suggesting strong dissolution of the glass during the experiments. A further increase in the amount of fluid produced the same contact morphology, but also increased the reactivity of glass. Complete recrystallization of the spheres was observed in experiment ENS2 conducted with 15 vol.% fluid (Fig. 3G).

Raman spectroscopy was used to check the possible incorporation of water in the glass spheres and newly formed material in the porosity. Raman spectra of the initial glass spheres indicated that they were water-free (Fig. 4Aa). In experiments with 1 vol.% fluid, very smooth peaks were detected in the spectral range of water (3500–3750 cm⁻¹), indicating that if any, only a very small amount of water was absorbed at the rim of the spheres (Fig. 4Aa). The Raman spectra of the material filling the porosity in experiments with 5 vol.% fluid showed well-defined bands at 3550-3750 cm⁻¹, 615 and 1100 cm⁻¹ (Fig. 4A), which are characteristic of a hydrous crystalline phase. From comparison of the Raman spectra, this material resembles a Ca-bearing zeolite (Fig. 5), even though it is Al-free. A profile from this recrystallized material towards and through a glass sphere shows that glass hydration and recrystallization also occur at the rim of the glass spheres (Fig. 4B). At the contact between two glass spheres, neither glass crystallization nor glass hydration was observed or detected (Fig. 4B). These observations suggest that glass hydrolysis was associated with recrystallization. Furthermore, the initial macroporosity disappeared and was replaced by a fluid-rich microporous recrystallized matrix. At the sphere contacts, the availability of water was much lower than on the free surfaces. Therefore, no water was incorporated in the glass and no recrystallization occurred.

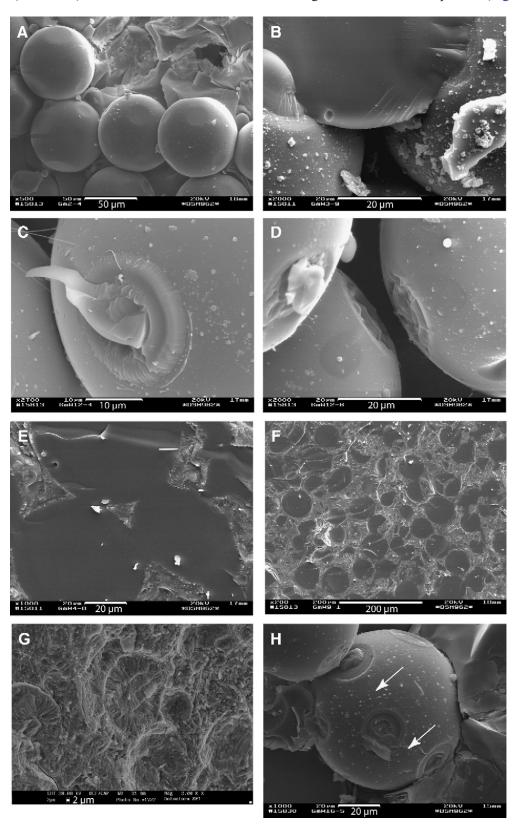
Regardless of the amount of fluid, evidence of cracking was found in some spheres in all experiments with the cracks spreading from one sphere contact to the next (Fig. 3H). Cracking increased significantly at low temperature and high confining pressure, but no relationship could be found between the amount of fluid and the amount of cracking. The timing of crack formation remains uncertain. Most cracks probably formed during the very first stage of the experiments during pressure loading, and at the end of the experiments during quenching. However, cracks might also have been formed during the runs.

The occurrence of all the features previously described indicates at least three main responses to stress:

ductile deformation in experiments performed with no or a small amount of fluid (≤ 1 vol.%), brittle deformation in all runs, and chemical reaction (dissolution-crystallization) in experiments with a large amount of fluid (≥ 5 vol.%).

3.2. Time-dependent change in strain and T, P, water content dependency

1-D strain was estimated from the change in diameter of the glass beads under compaction (Fig. 6A). Since the



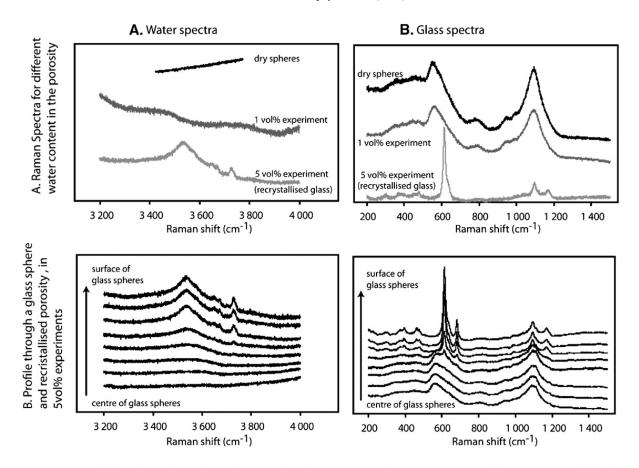


Fig. 4. Raman spectra within the glass spheres of experiments performed at 200 MPa and 350 $^{\circ}$ C. A.) Spectra of the water content within the material. B.) Spectra of the glass spheres and recrystallized material. A. Overview of the difference between experiments: in black an initial dry glass sphere (starting material), in grey a glass sphere from 1 vol.% fluid experiment (GmW1), and in light grey recrystallized material from the porosity of 5 vol.% fluid experiment (GmW4). B. Profiles over a distance of 32 μ m from within the glass spheres (bottom) through the recrystallized material (top) in the porosity of an experiment with 5 vol.% fluid (GmW4). The average spacing between successive measurement points is about 4 μ m. On all the graphs, the different spectra have been shifted vertically.

changes in the average sphere diameter due to compaction were negligible, the 1-D strain associated with each contact (ε , shortening of the sphere radius at the contact) can therefore be calculated using the following relation:

$$|\varepsilon| = \frac{d_{\rm s} - \sqrt{d_{\rm s}^2 - d_{\rm c}^2}}{d_{\rm s}} \tag{1}$$

where d_c is the contact diameter and d_s the associated sphere diameter directly measured for each sphere on the SEM pictures (Fig. 6A). The size of the glass spheres d_s and the contact size d_c were determined from SEM

pictures of the run products. Only unfractured and unbroken contacts were measured in order to estimate ductile strain. For each experiment, about 100 contacts were measured to calculate average contact diameters (N_c in Table 1), to ensure that the local strain measured at different grain contacts is representative of the overall strain of the aggregate. Since the confining pressure was isotropic, the strain was also assumed to be isotropic. Eq. (1) was only used to estimate the strain in dry experiments and in experiments with 1 vol.% fluid for $T \le 350$ °C, i.e., in experiments for which no glass recrystallization occurred. In experiments with 5 vol.% fluid or with 1 vol.% fluid at 450 °C, the material is not

Fig. 3. SEM-pictures showing the contact morphology in a glass sphere aggregate in experiments performed at 200 MPa and 350 °C. A. Flat contacts in dry experiments (Gm2). In experiments with 1 vol.% fluid, three types of contacts are observed: B. Contacts with bonding at the core and a flat rim (GmW3), C. Concentric rims of outgrowths and depressions (GmW12), and D. Stylolite-like contacts (GmW12). E. Filled porosity in 5 vol.% fluid experiments; a similar structure is also observed at 450 °C with 1 vol.% fluid (GmW4). F. Filled porosity in 5 vol.% fluid experiments. Microporous recrystallized material, interpreted as Ca-rich zeolites, completely fills the porosity (GmW9). G. Filled porosity in 14 vol.% fluid experiments (ENS2). Increasing the amount of water increases the degree of reactivity and recrystallization of the spheres. With 15 vol.% fluid (G), the spheres are entirely recrystallized, whereas with 5 vol.% fluid (E, F) recrystallization only occurs at the rims and in 1 vol.% fluid experiments (B, C, D, H) no recrystallization is evidenced. H. Initiation of fractures (arrows) that radiate from one contact to the others (GmW16).

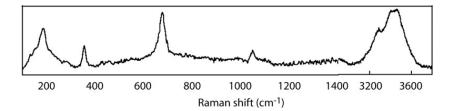
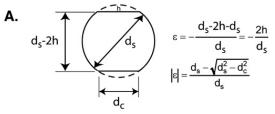


Fig. 5. Raman spectrum of a Ca-bearing Zeolite (thomsonite, from Wang et al., 1999).

loose and just a few sections of contacts could be observed with SEM. It was thus not possible to obtain a good statistical representation of the strain, which would thus be extremely badly constrained.

From Eq. (1) the finite strain at each contact was calculated at the end of the runs and the strain averaged over a significant number of contacts to obtain the



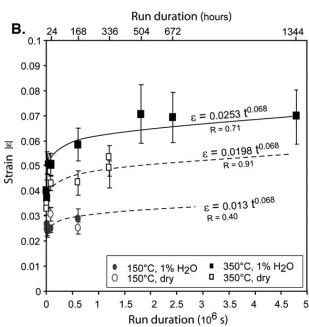


Fig. 6. A) Sketch to illustrate the method of calculating deformation at the grain contact using the initial sphere diameter (d_s) and the diameter of the truncated contact (d_c) . B. Time-dependent change in the average 1-D strain for dry (empty symbols) and 1 vol.% fluid (full symbols) experiments performed at 150 °C and 350 °C (P=200 MPa, circles and squares respectively). The curves represent mean-square regression fits of the experimental data following a power law. R is defined as the correlation coefficient using: $R = \frac{\text{cov}(X,Y)}{\sigma_X \sigma_Y}$, and $\text{cov}(X,Y) = \frac{1}{n} \sum_{j=1}^{n} (X_j - \overline{X})(Y_j - \overline{Y})$ where \overline{X} and \overline{Y} are the average values of X and Y and σ_X and σ_Y are the standard deviations.

overall strain achieved in each experiment (Table 1). Fig. 6B shows that in all experiments, most deformation occurred at the very beginning of the experiments, during the first 10 h. Then, deformation quickly slowed down after a few hours. Deformation-time relations were derived by combining the results obtained from different run-time experiments conducted under the same P-T conditions and fluid/solid ratio (Table 1, Fig. 6B). During the first 2 h of each experiment, a strong increase in sphere contact area was observed, which implies a strong decrease in stress and strain rate. For this reason, the $\varepsilon = f(t)_{[P,T]}$ relations were fitted with time-dependent power laws using only the strain data at t>2 h. A power law exponent value of 0.068 is best constrained by the 1 vol.% water experiments at 350 °C (Fig. 6B). In view of the wide scatter of the values obtained in all the experiments, this power exponent was left unchanged to fit the other experiments. For the 150 °C experiments, both the wet and dry experiments were fitted using the same $\varepsilon = \alpha t^{0.068}$ function (dashed line in Fig. 6B). Then, discrete differentiation of the ε -time relation data at given P and T values gives the microscopic strain rate (values listed in Table 1).

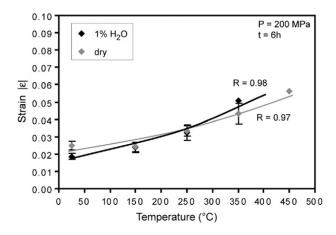


Fig. 7. Change in the average contact deformation of dry (grey diamonds) and 1 vol.% fluid (black diamonds) experiments versus temperature (P=200 MPa). The full lines represent mean-square regression fits of the experimental data using an exponential law. Error bars represent the standard deviation normalized to the average strain. R is defined as in Fig. 6.

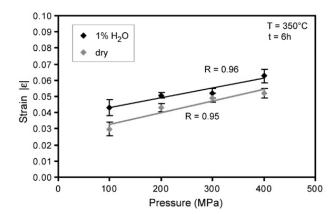


Fig. 8. Change in the average contact deformation of dry (grey diamonds) and 1 vol.% fluid (black diamonds) experiments versus confining pressure (T=350 °C). The full lines represent mean-square regression fits of the experimental data using a linear law. Error bars represent the standard deviation normalized to the average strain. R is defined as in Fig. 6.

At 350 °C, strain was clearly greater in the 1 vol.% fluid experiments than in the dry experiments, whereas the addition of water in the 150 °C experiments did not show any significant difference compared to the dry experiments. The strain versus time plots in Fig. 6B show that strain increased significantly with temperature. This can also be seen in Fig. 7, which shows a plot of the strain observed after 6 h in experiments conducted at a pressure of 200 MPa, and temperatures of 25, 150, 250, 350 and 450 °C. At $T \ge 250$ °C strain is enhanced by the addition of water. No quantitative data are available for the 1 vol.% fluid experiment performed at 450 °C. Indeed, as observed at 350 °C in experiments with 5 vol.%, the spheres chemically reacted with the fluid and the porosity was filled with new material.

A set of experiments with 6-hour run time was also performed at various confining pressures (100, 200, 300, 400 MPa) at 350 °C (Fig. 8). As expected from the

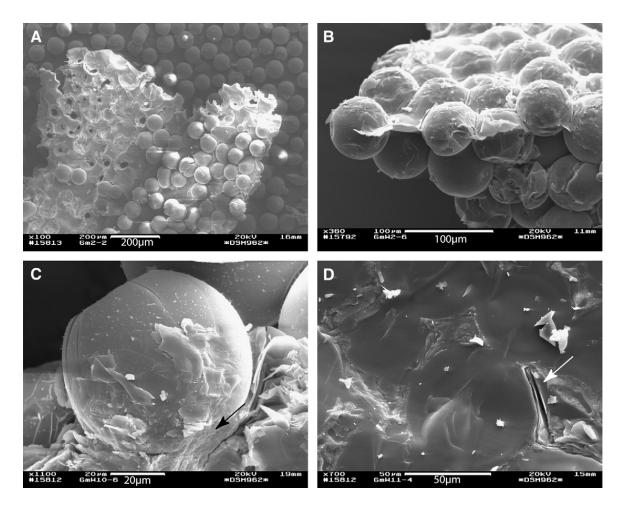


Fig. 9. Evidence of dissolution and indentation at the contact between glass spheres and mica from SEM-pictures. A) Mica mechanical indentation in a dry experiment (Gm2). B) Kinked mica and broken spheres in a 1 vol.% experiment (GmW2). C) Dissolution of a glass sphere at the contact with a mica flake in an experiment with 1 vol.% water (black arrow, GmW10). D) Dissolution of a glass sphere at the contact with a mica flake in an experiment with 5 vol.% water (white arrow, GmW11).

previous ε –T results after 6 h, experiments containing 1 vol.% fluid showed a higher strain than dry experiments at all pressures. Radial fracturing spreading from the contacts increases with confining pressure (Fig. 3H), so that the number of contacts between unbroken spheres is lower in experiments with high rather than low confining pressure. However, Fig. 8 shows that, for constant water content, the average strain increases linearly with increasing confining pressure.

All these observations indicate that strain increases with temperature, confining pressure, and the addition of even very small amounts of water. The similarity observed at 350 °C in 5 vol.% fluid experiments and at 450 °C in 1 vol.% fluid experiments, and the fact that, at 150 °C, dry and 1 vol.% fluid experiments have similar strain—time variations, suggest that the effect of water was thermally activated. In contrast, the effect of pressure was similar for the dry and 1 vol.% fluid experiments. The observed increase in strain was thus simply related to an increase in local stress at the contact between the spheres.

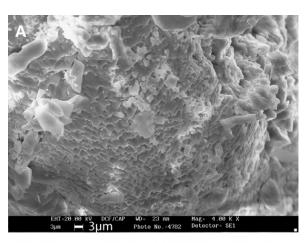
3.3. Effect of phyllosilicates

3 wt.% biotite was introduced in the starting mixture, in order to investigate whether or not the presence of mica enhances deformation as discussed in the literature (Rutter, 1983; Tada and Siever, 1989; Dewers and Ortoleva, 1995; Hickman and Evans, 1995; Bos et al., 2000; Rutter and Wanten, 2000; Renard et al., 2001; Niemeijer and Spiers, 2002). Glass sphere-mica contacts were scarce and difficult to identify because the samples lost their cohesion when opening the capsule. In dry experiments, the mica deformed mechanically by kinking and indentation (Fig. 9A). In experiments with 1 vol.% fluid, the mica was also kinked (Fig. 9B) and the spheres were slightly dissolved at their contacts (Fig. 9C). The mica could also be dissolved when a fine flake was sandwiched between two spheres. In experiments with 5 vol.% fluid, it was not possible to make any systematic direct observations of the mica because the mica flakes were closely sandwiched inside an aggregate of very compact recrystallized glass spheres. Some spheres were occasionally truncated at their contacts with mica (Fig. 9D), but it was hardly possible to quantify any strain or dissolution of these spheres. If at all, the dissolution of glass spheres at the contact with mica was low, and mica also dissolved.

Even though glass sphere dissolution was evidenced at the contact with the mica, it was impossible to quantify the amount of dissolution and the resulting strain related to the presence of mica. In conclusion, SEM observations indicate that the effect of mica on glass sphere dissolution was mostly limited to spheres directly in contact with the mica flakes.

3.4. Comparison with natural materials

In order to compare the results obtained for glass spheres with natural materials, several experiments were carried out using natural quartz under the same P-T and water content conditions as used with the glass beads (Table 1). The initial grains already presented some features of previous dissolution, but the surface of the grains was relatively smooth. After a few weeks, micro-stylolites developed at the contact between quartz grains, forming rough surfaces, while free faces only showed dissolution patterns (Fig. 10). These observations indicate that IPS was efficient under isostatic compaction even with very small amounts of water. However, the unknown and variable initial shape



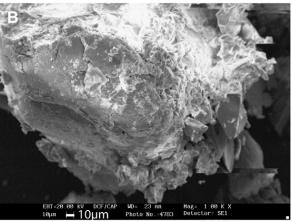


Fig. 10. Dissolution patterns on quartz grains (QmW10). A) Morphology of a contact between two quartz grains showing evidence of intergranular pressure solution (micro-stylolite). The small debris are produced by grain crushing under brittle deformation. B) View of the quartz aggregate and the previous contact.

of the quartz grains precluded any quantification of the strain of individual grains, as was done for the glass spheres.

4. Discussion

The aim of this study was to quantify the strain of the particles of an aggregate using a simple experimental design (isostatic confining pressure, constant temperature), for various P-T-t conditions and amounts of fluid. Deformation processes and the heterogeneity of the local stresses are discussed first. The experimental results are then compared with glass creep according to an exponential law and with the creep of natural minerals according to a power law. Finally, the implications of this experimental study are discussed for several geological fields where fluid-enhanced deformation under high P-T conditions is relevant.

4.1. Deformation processes

Experiments were performed under P-T conditions suitable for IPS deformation in natural silica-rich rocks, but the experimental results obtained indicate that at least three main deformation processes occurred in the simple system used, involving only glass spheres \pm 3 wt.% mica and pore water:

- *Brittle deformation* (cracking) occurred in all experiments, but the level of deformation depends primarily on the confining pressure and to a lesser extent on temperature.
- Ductile deformation was also observed in all experiments, illustrated in the dry experiments by the development of flat contacts due to plastic flow. This deformation process is very rapid and occurred at the very beginning of all experiments. Fig. 7 shows that plastic flow was thermally activated, which is consistent with a decrease in effective glass viscosity with increasing temperature (Fig. 1). Plastic flow also occurred in the experiments conducted with water, as suggested by the flat outer rings bordering the contacts between the spheres in these experiments. Although glass hydration was not evidenced by Raman spectroscopy in the 1 vol.% fluid experiments, the increase in strain with the addition of water is likely due to the decrease in effective glass viscosity resulting from partial hydration of the glass. Water affects glass by decreasing the connectivity of the network according to the following chemical reaction:

$$-Si-O-Si- + H2O \rightarrow -Si-OH HO-Si-$$
 (2)

This reaction leads to the replacement of covalent chemical bonds by hydrogen bonds, thereby inducing a reduction in the effective glass viscosity. The observed enhancement of recrystallization with 1 vol.% fluid at T>350 °C suggests that the hydration reaction of glass was thermally activated. However, another deformation process occurred in the experiments with water. Given the shape of the inside surface of the contacts and the fact that spheres were often bonded together it would seem that not all the water entered the glass spheres, and that fluid-driven deformation, such as IPS, might also have been active in the 1 vol.% water experiments. Fluid-driven deformation was clearly less efficient than plastic flow at the beginning of the experiments. However, it became the dominant deformation mechanism in experiments lasting more than 2 weeks, once the contact between the spheres reached a critical size above which the local stress was too low for plastic deformation. In any case, the low amount of fluid was not a limiting factor for IPS, since it proved to be effective in the quartz bearing experiments with 1 vol.% water. The observations made in these experiments also suggest that, in the presence of a small amount of water, there is a critical contact size above which sintering occurred. This effect strengthened the aggregate and acted against further ductile deformation.

- Chemical solid-fluid reactions occurred at 450 °C even for low fluid content and at 350 °C in the 5 vol.% fluid experiments. Under these conditions, the glass spheres crystallized and also dissolved on their free faces. This is reflected by the occurrence of reaction rims on free faces and by the presence of a new phase in the porosity (Fig. 3E-G). Assuming that the initial porosity of the aggregates was about 32% (based on a body-centered cubic packing), 5 vol.% and 1 vol.% of water filled respectively 14% and 3% of the porosity. The extent of the reaction was surprisingly high in view of the low amount of fluid involved in the experiments and considering that the transport of elements from the spheres toward the porosity has to take place by diffusion in a thin fluid layer confined at the grain contacts. Solid-fluid chemical reactions evidenced at 350 °C in 5 vol.% experiments and at 450 °C in 1 vol.% experiments thus indicate that dissolution and precipitation kinetics, as well as transport kinetics in the fluid layer, were very high. The difference in experimental conditions between experiments showing almost no precipitation in the porosity (1 vol.% fluid at 350 °C) and those showing a porosity almost completely filled with newly crystallized products (5 vol.% fluid at 350 °C and 1 vol.% fluid at 450 °C) indicates that the

extent of fluid-glass interactions was closely dependent on both temperature and amount of fluid.

4.2. Heterogeneity of local stress in an aggregate

Local normal stresses applied at the sphere contacts (σ_c) at the end of the experiment were calculated from the size of the contact and the applied confining pressure using:

$$\frac{\sigma_{\rm c}}{d_{\rm s}^2} = \frac{4\sigma}{\pi d_{\rm c}^2} \tag{3}$$

where σ is the confining pressure (MPa), d_s the sphere diameter (m) and d_c the contact diameter (m). Local stresses were calculated for each contact and then averaged over the sample. According to the strain estimates given in Table 1, porosity changes related to compaction were less than 2% of the initial porosity, so that changes in fluid pressure during compaction can reasonably be neglected. Furthermore, in experiments with 5 vol.% fluid with an initial 30% porosity, the pore pressure developed up to 350 °C would be only up to a maximum of about 10 MPa, whereas at 450 °C, under supercritical conditions it would be about 40 MPa. Consequently, the fluid pressure is much lower than the local stress at the grain contacts (Table 1) and can therefore be neglected when estimating the effective stress at grain contacts according to Eq. (3).

Except for experiments performed at 400 MPa, cracking was only evidenced at a few contacts and was therefore observed to be a secondary deformation mechanism. Eq. (3) is thus only suitable for calculating the local normal stress at plastically deformed contacts that were not fractured.

In all experiments, the dispersion of contact diameters is 2 to 7% of the average diameter (Table 1), showing that the strain was heterogeneously distributed at grain scale in the samples. The calculated local stress dispersion is 5 to 30% of the averaged local stress, with most experiments having a lower heterogeneity with a dispersion of 5 to 15%. Even if these heterogeneities were observed at small scale, they average in the volume and the deformation at the sample scale was homogeneous. Consequently, using a sufficient number of microscopic measurements, a macroscopic constitutive flow law could be calculated for the aggregate.

4.3. Comparison with creep of glass

The relative contributions of the three deformation mechanisms mentioned in Section 4.1 may have

changed during the experiments, but they could have also been active at the same time. However, the results obtained indicate that strain was mostly accommodated by plastic flow, at least in the dry experiments and during the first stage of the experiments with added water. Plastic deformation in glass cannot occur by the movement of dislocations as in crystallized material. Instead, deformation occurs by the rearrangement of groups of 10 to 100 atoms. The macroscopic strain rate $\dot{\epsilon}$ of plastic deformation in an amorphous material can be described by the following equation (Spaepen, 1981; Heggen et al., 2004):

$$\dot{\varepsilon} = 2c_{\rm f}k_{\rm f}\frac{B}{\Omega}\sinh\left(\frac{B}{2kT}\sigma\right) \tag{4}$$

where σ is the external pressure, $k_{\rm f}$ is a rate constant, Ω is the atomic volume, $c_{\rm f}$ is the concentration of defects, k is the Boltzmann constant, T is the temperature, and B is the activation volume. When $\frac{B}{2kT}\sigma$ is high, $\left(\frac{B}{2kT}\sigma\right)$ can be simplified to $\frac{1}{2}\exp\left(\frac{B}{2kT}\sigma\right)$ and Eq. (4) can be rewritten as a more classical exponential law.

$$\dot{\varepsilon} = AB \, \exp\left(\frac{B}{2kT}\sigma\right) \tag{5}$$

with
$$A = \frac{2c_f k_f}{\Omega}$$
.

A plot of $\dot{\epsilon}$ as a function of the local normal stress at the contact (Fig. 11) shows a linear trend on a linear–log scale. Therefore, the experimental data can be reasonably explained by an exponential law (Eq. (5)). The slope of the $\ln \dot{\epsilon} - \sigma$ lines in Fig. 11 corresponds to the values of B/2kT, which are respectively $2.69 \pm 1.16 \times 10^{-9}$ and $3.64 \pm 1.96 \times 10^{-9}$ for dry and wet experiments at 350 °C. From these values, activation volumes $B=46\pm20$ ų and $B=63\pm32$ ų are obtained for the dry and wet experiments (Table 2). In view of the very wide scatter of the $\ln \dot{\epsilon} - \sigma$ experimental data at 150 °C, it was not possible to calculate an activation volume from these data, so it was assumed that it was the same as at 350 °C.

The pre-exponential factors derived from Fig. 11 for both the dry and wet experiments at 350 °C are: $AB_{\rm dry}=1.91\pm1.32\times10^{-12}~{\rm s}^{-1}$ and $AB_{\rm 1}~{\rm vol.\%}=4.16\pm7.06\times10^{-13}~{\rm s}^{-1}$. For the 150 °C experiments, a value of $AB=5.69\pm41.03\times10^{-15}~{\rm s}^{-1}$ was obtained for both the dry and wet experiments, when using $B=46\pm29~{\rm \AA}^3$. The increase in AB with temperature corresponds to an increase in A from about $10^{-16}~{\rm \AA}^{-3}~{\rm s}^{-1}$ at 150 °C to about $10^{-14}-10^{-15}~{\rm \AA}^{-3}~{\rm s}^{-1}$ at 350 °C in both the wet and dry experiments respectively. These variations quantify the joint effect of temperature and water on the rate constant $k_{\rm f}$ in Eq. (4).

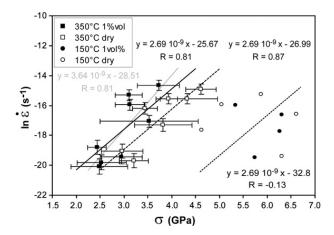


Fig. 11. Strain rate (s⁻¹) versus contact (local) stress (GPa) for both dry (empty symbols) and 1 vol.% experiments (full symbols) performed at 150 °C and 350 °C (circles and squares respectively) using an exponential deformation law (ln \dot{e} versus σ , from Eq. (5)). The black lines represent exponential fits of all experiments assuming a constant slope for B=46 ų (see text for more explanations). The full lines represent fits of 1 vol.% fluid experiments at 350 °C, the dashed line the fit of dry experiments at 350 °C, and the dotted line the fit of both dry and 1 vol.% fluid experiments at 150 °C. The grey line is the best exponential fit of the experiments performed with 1 vol.% fluid (B=63 ų). The fits and regression equations are given in SI units (σ in Pa). R is defined as in Fig. 6.

Fig. 11 gives B_1 vol.%> $B_{\rm dry}$ and A_1 vol.%< $A_{\rm dry}$ (Table 2). The presence of fluid in the porosity and in the contact zone between two glass spheres is likely to affect the rate constant $k_{\rm f}$, while $c_{\rm f}$ and Ω remain constant. Therefore, Fig. 11 suggests that $k_{\rm f}$ vol.%< $k_{\rm f}$ dry. However, these experiments indicate that strain and strain rate are lower in dry experiments than in experiments performed with 1 vol.% fluid, suggesting that the rate constant $k_{\rm f}$ is larger in the wet experiments. Assuming that the activation volume B is constant in both the dry and wet experiments implies that A_1 vol.%> $A_{\rm dry}$ and $A_{\rm f1}$ vol.%> $A_{\rm fry}$. Indeed, if as shown in Fig. 11, the best fit of the ln $\dot{\varepsilon}-\sigma$ data obtained at 350 °C in the wet experiments is plotted using a fixed B=46 ų (as obtained from the dry experiments), AB_1 vol.%=

 $7.11\pm4.91\times10^{-12}$, which corresponds to $A_{1 \text{ vol.\%}}=1.54\pm1.06\times10^{-13} \text{ Å}^{-3} \text{ s}^{-1}$. This value of A is now higher than that obtained for the dry experiments $(A_{\text{dry}}=4.06\times10^{-14} \text{ Å}^{-3} \text{ s}^{-1})$, which is consistent with an increase in the rate constant k_{f} resulting from glass hydration. It is therefore likely that the activation volumes of dry and hydrated glass are similar.

An interesting result is that the effective glass viscosity estimated as $\eta_{\rm eff}=\sigma/\dot{\epsilon}$ lies in the range $10^{15}-10^{18}$ Pa s. These values are in agreement with the extrapolation of the viscosity of soda-lime glass measured at high temperature to the temperature of the experiments described here (Fig. 1). Furthermore, the effective glass viscosity is higher in dry than in wet experiments and in the 150 °C experiments than in those at 350 °C (Table 1).

4.4. Comparison with power-law creep

Deformation of crystalline material at moderate temperature involves dislocation creep, which is well described by a power law. As mentioned above, glass deformation does not involve the displacement of dislocation and may be described by exponential creep laws. However, in view of the large variation in strain rate with varying contact stresses (more than a factor 100 increase in $\dot{\epsilon}$ is observed when σ is doubled, see Fig. 11) and experimental data scatter, it is reasonable to describe these experiments using a power law relationship. Such a constitutive law is widely used for rocks subjected to stress and strain rates corresponding to those of the crust (Poirier, 1985):

$$\dot{\varepsilon} = A_1 \exp\left(-\frac{E}{RT}\right) \sigma^n \tag{6}$$

with A_1 and n being the power law parameters (A_1 in Pa⁻ⁿ s⁻¹), E the activation energy (kJ mol⁻¹), R the gas constant (kJ mol⁻¹ K⁻¹) and T the temperature (K). The Arrhenius term accounts for the increase in strain rate

Table 2 Deformation law parameters

		Eq. (5)			Eq. (6)							
		B (Å ³)	$A (Å^{-3} s^{-1})$	$AB (s^{-1})$	E (kJ/mol)	$A_1 (\operatorname{Pa}^{-n} \operatorname{s}^{-1})$	$A_1 \exp(-E/RT)$	n				
350 °C	Dry	46±20	$4.12\pm1.07\times10^{-14}$	$1.91 \pm 1.32 \times 10^{-12}$	78±45	$4.82 \pm 42.16 \times 10^{-102}$	$1.50 \pm 1.04 \times 10^{-108}$	10.5±2.2				
	1 vol.% H ₂ O	63 ± 32	$6.65 \pm 7.87 \times 10^{-15}$	$4.16\pm7.06\times10^{-13}$	96 ± 45	$8.88 \pm 77.73 \times 10^{-100}$	$8.01 \pm 5.54 \times 10^{-108}$	10.5 ± 2.2				
	1 vol.% H ₂ O	$46\!\pm\!20$	$1.54 \pm 1.06 \times 10^{-14}$ (*)	$7.11 \pm 4.91 \times 10^{-12}$								
150 °C	Dry		$1.23\pm3.67\times10^{-16}$ (*)									
	1 vol.% H ₂ O	$46\!\pm\!56$	$1.23 \pm 7.37 \times 10^{-16}$ (*)	$5.69 \pm 20.59 \times 10^{-15}$								

^(*) Values obtained by a best-fit of the $\ln(d\varepsilon/dt) - \sigma$ experimental data with a fixed *B*. Uncertainties are calculated as 2Σ with Σ the standard deviation.

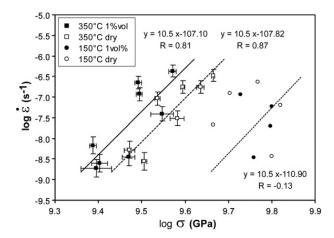


Fig. 12. Strain rate (s⁻¹) versus stress (Pa) for both dry and 1 vol.% experiments performed at 150 °C and 350 °C following a power law (ln $\dot{\epsilon}$ versus ln σ , from Eq. (6)). The full and dashed lines represent the best exponential fits of 1 vol.% fluid experiments and dry experiments, respectively. These two lines have the same slope (n=10.5). Experiments performed at 150 °C are fitted using n=10.5 as for the 350 °C experiments. Dry and 1 vol.% experiments are fitted with the same power law. The fits and regression equations are given in SI units (σ in Pa). R is defined as in Fig. 6.

with temperature (Fig. 7). The validity of the power law for the conditions of the experiments described here is demonstrated in Fig. 12, which shows that the $\log(\dot{\varepsilon}) - \log(\sigma)$ data for the different sets of experiments can be reasonably fitted by a straight line.

The stress exponent and pre-exponential factors in Eq. (6) can be obtained from the slope and the intercept of these plots. Fig. 12 shows that the dry and wet experiments can be fitted with the same stress exponent, $n=10.5\pm2.2$ at 350 °C. These values are much higher than those generally obtained for crystalline material deformed by dislocation creep at similar conditions, n=2to 4 for quartzite (Kronenberg and Tullis, 1984 and references therein; Poirier, 1985; Koch et al., 1989) and n=5-8 for salt (Poirier, 1985). However, the high stress exponent deduced from the experiments is in reasonable agreement with those reported by Kingery et al. (1976) for glass (from 4 to 20) or for plasticine at room temperature (up to n=7.3; Zulauf and Zulauf, 2004). Kronenberg and Tullis (1984) evidenced a decrease of the stress exponent with fluid availability, whereas this study shows that both dry and wet experiments have similar stress exponent with similar uncertainties. However, one should note that the pre-factor $A_1 \exp(-E/RT)$ of the power-law creep increases 5 times with the addition of water, from $1.50\pm1.04\times10^{-108}~Pa^{-10.5}~s^{-1}$ in dry experiments to $8.01 \pm 5.54 \times 10^{-108} \text{ Pa}^{-10.5} \text{ s}^{-1}$ in wet experiments. This evolution shows the strong enhancing effect of water on deformation rate.

The activation energy can be calculated using the intercepts $(A_1 \exp(-E/RT))$ in Eq. (6) of the $\log(\dot{\varepsilon}) = f$ $(\log(\sigma))$ fits obtained for the 150 °C and 350 °C experiments with the same stress exponent. Since the same $\varepsilon = f(t)_{[P,T]}$ function was used to fit the dry and wet experiments at 150 °C (dashed line in Fig. 12), different activation energies are obtained when using the dry or wet experiments at 350 °C. Values of $E=78 \text{ kJ mol}^{-1}$ and 96 kJ mol⁻¹ were obtained when using the dry and wet experiments respectively, assuming A_1 is independent of temperature. A_1 calculated values are thus $4.82\pm42.16\times10^{-102}~\mathrm{Pa}^{-10.5}~\mathrm{s}^{-1}$ for dry glass and $8.88\pm77.73\times10^{-100}~\mathrm{Pa}^{-10.5}~\mathrm{s}^{-1}$ for hydrated glass. For plastic processes such as dislocation creep, the availability of water accounts for a decrease in activation energy from 300 kJ mol⁻¹ in vacuum-dried quartzite to $220-170 \text{ kJ mol}^{-1}$ in "as is" samples down to 120-150 kJ mol⁻¹ in wet quartzites, and for a variable increase in the pre-exponential parameter A_1 (Kronenberg and Tullis, 1984 and references therein). The increase in the pre-exponential parameter A_1 observed in this study is thus consistent with observations of dislocation creep of quartzite (Kronenberg and Tullis, 1984 and references therein; Koch et al., 1989). Activation energies estimated here for plastic deformation of amorphous materials are 1.5-2 times lower than those estimated for crystalline materials, but they are similar to those of intergranular pressure-solution (73 kJ mol⁻¹ Dewers and Hajash, 1995). The increase of the activation energy from wet to dry experiments is opposite to the variations reported in the literature for crystalline material. However, in view of the large uncertainties on the estimated activation energies, this apparent variation is not significant.

4.5. Comparison with natural materials

Our experiments with glass spheres indicate a complex deformation process involving mostly plastic flow and fluid-enhanced deformation (hydrolytic weakening). This study highlights the difference between the rheology of amorphous and crystalline materials on one hand, and the strong effect of a free-fluid phase in rock softening on the other hand. It indeed provides new insights on the rheology of soda-lime glass, which is found in glass-bearing rocks, but which is also used for analogue experiments. Our experiments are thus relevant for the study of fluid—rock interaction coupled with deformation, especially in glass-bearing rocks, where few data are available in the literature.

Rock softening due to the presence of fluid has been largely evidenced in the continental crust in ductile shear zones where fluid circulation coupled with highgrade deformation changes the rock rheology (Boullier and Guéguen, 1975; Behrmann and Mainprice, 1987; Martelat et al., 1999; Zulauf et al., 2002; Rosenberg and Stünitz, 2003). Many experimental studies were performed during the last decades in order to quantify diffusion rates and diffusion length as well as changes in rheology related to hydrolytic weakening at mid-crustal P-T conditions (e.g., Griggs and Blacic, 1964; Kronenberg and Tullis, 1984; Paterson, 1989; Kronenberg, 1994 for reviews). At higher temperatures, experiments show that plastic flow is enhanced by the presence of a few percent of melt (Kohlstedt and Zimmermann, 1996; Rosenberg, 2001 for reviews). In partially molten rocks, films of melt localized within the grain contact enhance the diffusion of dissolved material to the pores, a process equivalent to IPS. In our experiments, the presence of water in the porosity has the same effect as the presence of small amounts of melt in the porosity of HT rocks (<20% melt). The simple low-T experiments on glass spheres described here confirm that the presence of fluid increases plasticity. They can be considered as a first order analogue for the macroscopic study of silicate deformation at higher P-T conditions. The use of glass spheres may thus allow analogous melt experiments to be performed using a simple experimental design as well as experiments at lower P-T conditions than usual melt experiments. Compared to other experiments, the main advantage of our experimental set-up is that the initial aggregate geometry can be deconvoluted from the deformation state.

Many studies have been performed in order to understand the rheology of crystalline rocks at various P-T conditions. However, there are much fewer studies investigating the rheology of glass-bearing rocks, and the effect of a glassy phase in crystalline rocks. Our experiments show that for P-T-t conditions favourable for IPS in crystalline rocks, amorphous materials deform plastically. Glass-bearing rocks are thus expected to deform in a more ductile way than holocrystalline rocks. At 1 GPa confining pressure and a strain rate of 10⁻⁶ s⁻¹, Hacker and Christie (1991) observed ductility at 675 °C in glassy basalt, instead of 800 °C in diabase (Kronenberg and Shelton, 1980). In addition to extending the ductile domain to lower temperature, the presence of glass also switches the mechanism of strain accommodation: glassy basalt accommodates ductile deformation by plastic flow within the glass, and not by dislocation glide and twinning as observed in diabase (Hacker and Christie, 1991). Such a difference is not only related to high-temperature and high pressure rocks. Recent experiments show that

glassy basalt is much weaker than diabase at low temperature (500 °C, 5 MPa; Keulen, 2006). By investigating glass deformation at mid-crustal P-T conditions, our experimental study provides new and direct insights on the rheology of soda-lime glass, and indirect information on the rheology of glassy rocks and on the competition of various deformation mechanisms in aggregates containing grains and materials with different plastic limits. A possible extension of our experiments could thus be to deform an aggregate containing both glass spheres and quartz grains in various proportions.

Soda-lime glass has been recently used in analogical experiments to model the deformation of pyroclastic material (Quane et al., 2004; Quane and Russell, 2005, 2006; Roche et al., 2005). Quane and Russell (2005, 2006) conducted compaction experiments with glass beads under constant uniaxial strain rate as an analogue for the welding of pyroclastic material. However, they used dry aggregates and did not investigate the effect of pore fluid during compaction. Sparks et al. (1999) suggested that absorption of water from trapped interstitial fluids in ignimbrite glass would enhance welding by reducing the ignimbrite's viscosity, which is confirmed by our experiments. Even though our experiments were performed at different P-T conditions, by investigating the effect of water on welding, they are complementary to those of Quane and Russell (2005, 2006), and are the link between their experiments and the observations of Sparks et al. (1999).

5. Conclusion

Isostatic compaction experiments were performed on quartz and glass sphere aggregates under mid-crustal P–T conditions (200 MPa and 350 °C) in order to quantify grain-scale strain and to make a comparison with theoretical models of ductile deformation in granular analogues of rocks. The main advantage of this approach is that, using low-cost pressure vessels, more than sixty deformation experiments could be performed. This made it possible to test a wide range of confining pressures, temperatures, and amounts of fluid.

Under the experimental P–T conditions, quartz and glass spheres do not deform by the same process. Natural materials show evidence of intergranular pressure-solution (IPS), while experiments with glass spheres show evidence of both, brittle and ductile deformation processes. Even though experiments using glass spheres show little evidence of IPS, ductile strain by grain contact

plastic flattening can be significant in systems subjected to isostatic pressure conditions. However, IPS is an efficient deformation mechanism in the quartz aggregates experiments in the presence of 1 to 3 vol.% water and in the absence of strong external deviatoric stress.

Dry experiments show that glass spheres accommodate stress concentrations by plastic strain. The addition of water in the porosity allows the local stresses to be accommodated by additional deformation mechanisms. The experiments described show that fluid-glass interactions can be very high, even with very low amounts of fluid at T>250 °C. Below this temperature, the influence of water on strain decreases rapidly, and no influence was found for long-duration experiments lasting up to 2 weeks at 150 °C with a low fluid content. In contrast, a very high reactivity of glass with water was observed at 450 °C, leading to the precipitation of new material filling all the porosity. The same observations were also made at 350 °C, but in the presence of a larger (but still very low) amount of water. With a large amount of fluid, chemical processes are dominant due to very rapid kinetics of dissolutiontransport-precipitation of the glass.

In dry and wet experiments, deformation of a glass sphere aggregate is accommodated mostly by plastic flow that can be described by either an exponential or a power law for the range of experimental conditions. With power-law creep, the presence of a small amount of water does not affect the stress exponent (n=10.5 in both the dry and 1 vol.% experiments at 350 °C), but leads to a decrease in the pre-factor.

This study clearly highlights the key effect of water, of glassy materials, and to a lesser extent temperature, in rock softening. The simple experiments presented in this study can thus be used as a first approximation of natural silicate rocks deformed in the presence of fluid at high pressure and temperature. They give direct insights on the rheology of soda-lime glass and by extension of glass-bearing rocks under mid-crustal P-T conditions.

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